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MODIFICATION OF THE STRUCTURE OF AMORPHOUS DIRECTIONAL  
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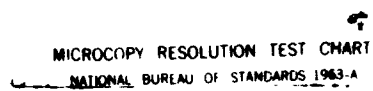
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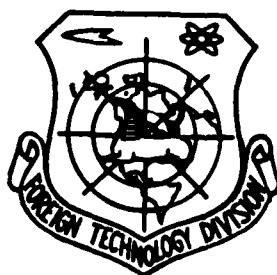
## FOREIGN TECHNOLOGY DIVISION



MODIFICATION OF THE STRUCTURE OF AMORPHOUS DIRECTIONAL POLYMETHYL  
METHACRYLATE BY METHODS OF GRAFTED AND BLOCK POLYMERIZATION

by

A.I. Kurilenko, V.P. Yakimtsov, L.P. Krul'



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By: A.I. Kurilenko, V.P. Yakimtsov, L.P. Krul'

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Foreign page numbers occur in the English text and may be found anywhere along the left margin of the page as in this example:

In them occurs the state named "night blindness" - hemeralopia, which, according to the current point of view, is a result of damage of the rod-shaped apparatus of the eye.

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However, in recent years it has been shown that with the hereditary pigment degenerations in animals the biochemical changes are observed in all cellular elements of the retina.

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# U. S. BOARD ON GEOGRAPHIC NAMES TRANSLITERATION SYSTEM

Block	Italic	Transliteration	Block	Italic	Transliteration
А а	<i>А а</i>	A, a	Р р	<i>Р р</i>	R, r
Б б	<i>Б б</i>	B, b	С с	<i>С с</i>	S, s
В в	<i>В в</i>	V, v	Т т	<i>Т т</i>	T, t
Г г	<i>Г г</i>	G, g	У у	<i>У у</i>	U, u
Д д	<i>Д д</i>	D, d	Ф ф	<i>Ф ф</i>	F, f
Е е	<i>Е е</i>	Ye, ye; E, e*	Х х	<i>Х х</i>	Kh, kh
Ж ж	<i>Ж ж</i>	Zh, zh	Ц ц	<i>Ц ц</i>	Ts, ts
З з	<i>З з</i>	Z, z	Ч ч	<i>Ч ч</i>	Ch, ch
И и	<i>И и</i>	I, i	Ш ш	<i>Ш ш</i>	Sh, sh
Й й	<i>Й й</i>	Y, y	Щ щ	<i>Щ щ</i>	Shch, shch
К к	<i>К к</i>	K, k	Ъ ъ	<i>Ъ ъ</i>	"
Л л	<i>Л л</i>	L, l	Ы ы	<i>Ы ы</i>	Y, y
М м	<i>М м</i>	M, m	Ь ь	<i>Ь ь</i>	'
Н н	<i>Н н</i>	N, n	Э э	<i>Э э</i>	E, e
О о	<i>О о</i>	O, o	Ю ю	<i>Ю ю</i>	Yu, yu
П п	<i>П п</i>	P, p	Я я	<i>Я я</i>	Ya, ya

\*ye initially, after vowels, and after Ъ, Ь; e elsewhere.  
When written as ё in Russian, transliterate as yё or ё.

## RUSSIAN AND ENGLISH TRIGONOMETRIC FUNCTIONS

Russian	English	Russian	English	Russian	English
sin	sin	sh	sinh	arc sh	$\sinh^{-1}$
cos	cos	ch	cosh	arc ch	$\cosh^{-1}$
tg	tan	th	tanh	arc th	$\tanh^{-1}$
ctg	cot	cth	coth	arc cth	$\coth^{-1}$
sec	sec	sch	sech	arc sch	$\operatorname{sech}^{-1}$
cosec	csc	csch	csch	arc csch	$\operatorname{csch}^{-1}$

Russian      English

rot      curl  
lg      log

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Page 412.

MODIFICATION OF THE STRUCTURE OF AMORPHOUS DIRECTIONAL POLYMETHYL METHACRYLATE BY METHODS OF GRAFTED AND BLOCK POLYMERIZATION.

A. I. Kurilenko, V. P. Yakimtsov, L. P. Krul'.

(Presented by academician P. A. Reh binder 30 September 1971).

Are in detail studied chemical aspects of problem of modification of properties of oriented crystalline polymeric materials by methods of synthesis in them of grafted and block copolymers and it is established that graft polymers are formed in amorphous zones of materials; however, mechanism of their effect on structure and mechanical properties of material is not clear [1, 2]. To model these processes is convenient in the amorphous polymeric materials, in which the oriented structures are not fixed/recorded by crystals, and therefore are more sensitive to any effects; the results of studying the processes of the grafted copolymerization in them and its effect on the structure and the properties of materials to more easily analyze. However, in the literature there is no information even about attempts at the realization of the grafted copolymerization in the oriented amorphous materials, probably, because the first stage of process - the sorption of monomer by material - leads to the relaxation of the oriented structures. By the analysis of the indirect data about the properties of the grafted materials is substantiated the mechanism of "cross-linking" of materials by the structures of grafted chains [3]. Consequently, it is possible to



select conditions, with which the relationship/ratio of the rates of the synthesis of graft polymer (PP) and sorption of monomers provides the effective cross-linking of the oriented amorphous materials earlier than in them the concentration of monomer, sufficient for the plastification, will be accumulated. The realization of this assumption could be the direct proof of the correctness of the assumed mechanism of cross-linking and its universality and, furthermore, would reveal the possibility of amorphous materials by the methods of the grafted copolymerization.

In this work processes of grafted and block polymerization of polyacrylonitrile (PAN) and polyvinylidene chloride (PVDKh) in amorphous films of atactic polymethyl methacrylate (PMMA), oriented by extract, are studied. The selected polymers possess the combination of the properties, necessary for checking the assumptions presented. Oriented PMMA has heterogenic fibrillar structure [4] (as crystalline materials) and is readily soluble in AN and VDKh. PVDKh in the process grafting is crystallized and it reproduces well the oriented structures of the materials of any nature [6], PAN - is cross-linked during the heating, both more heat- and temperature-resistant, than PMMA, and they are not dissolved in the monomers.

To graft they conducted from gaseous phase by method of straight/direct generation of free radicals ( $\gamma$ -rays  $\text{Co}^{60}$ , rate of dose of 150 rad/s,  $T=25^\circ$ ) in samples, placed into medium of AN and VDKh, and by method of post-polymerization (samples No's 6, 7). Pressure of

AN and VDKh was selected so that the rates of synthesis would be close ones. The structure of samples studied by methods of IKS [expansion uncertain; may be infrared spectrophotometry] in polarized light (R - the value of dichroic relation for  $\nu$  536  $\text{cm}^{-1}$  - PVDKh, 2248  $\text{cm}^{-1}$  - PAN, 752  $\text{cm}^{-1}$  - PMMA), isometric heating [5] and DTA [differential thermal analysis].

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The quantity of PAN and PVDKh ( $\Delta P$ ) they are expressed in the percentages of weight of PMMA. The results of investigations are given in Table 1 and Fig. 1 and 2.

Grafted PMMA retains oriented structures and possesses properties of three dimensional polymers independent of nature and conditions of synthesis of graft polymer, which is evident from results of studying of solubility (at 20°) and thermomechanical properties.

In dichloroethane (solvent PMMA) samples No 2 and No 4 became stratified (isolated layers they are designated as samples No's 3 and 5), No 6 only will swell, and from No 7 is extracted ~30% PMMA. In the dimethyl formamide (solvent PMMA and PAN) samples No's 2, 5, and 7 are dissolved completely, but after heating during 15 min. at 160° they behave just as in the dichloroethane. With heating the fixed/recorded along the length samples in them are developed and increase with an increase in the temperature the internal stresses/voltages, caused by the entropy elasticity of the defrosted

oriented structures of PMMA [7]. Maximum stresses/voltages  $\sigma_{\max}$  in initial PMMA are observed when  $T_{\max} = 80^\circ$ , further temperature rise causes decrease  $\sigma$  as a result of relaxing the structures and at  $133^\circ$  sample disintegrates. In grafted samples with  $\sigma_{\max}$  higher than in the original and when  $T > T_{\max}$  is maintained at the same level to  $T_s' = 150^\circ$  (samples No's 5, 6). Sharp decrease  $\sigma_s$  occurs only when  $T_s = 190-245^\circ$ , the samples with PAN retain their form and small  $\sigma$ , that grow at  $350-400^\circ$ .

From results of these experiments it unambiguously follows that grafted PMMA possesses structure of oriented three dimensional polymers, in which as the cross-links serve structures (blocks) of PAN and PVDKh, chemically combined with macromolecules of PMMA and fixing them until blocks are softened by heating or solvents. PAN during the heating chemically is cross-linked; therefore blocks of PAN do not disintegrate.

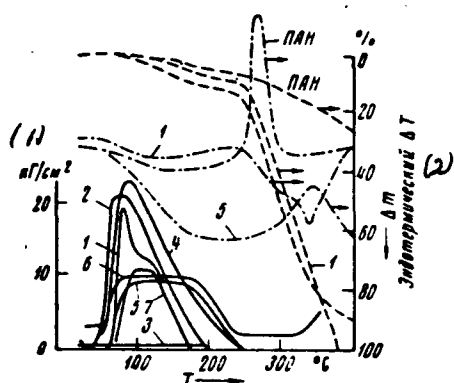


Fig. 1.

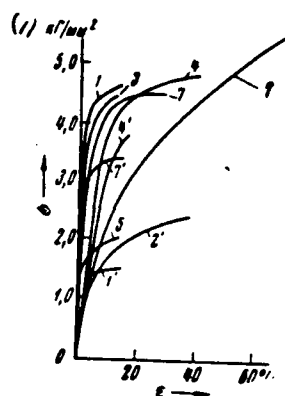


Fig. 2.

Fig. 1. Heat- and heat resistance of grafted films. Numerals in the curves correspond to the numbers of samples in Table 1. Diagrams of isometric heating - solid lines, thermogravimetric curves - dash ( $\Delta m$  - the decrease in weight of sample); thermogram - dot-and-dash lines.

Key: (1).  $\text{kgf/cm}^2$ . (2). Endothermal  $\Delta T$ .

Fig. 2. Stress-strain diagrams of samples at  $20^\circ$ . 1, 2, 4, 7 - to the heating, 1', 2', 4', 7' - after heating at  $160^\circ$ , 15 min.

Numerals in the curves correspond to the numbers of samples in Table 1.

Key: (1).  $\text{kg/mm}^2$ .

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Since PMMA is destroyed in a field of  $\gamma$ -radiation [8] and its oriented structures relax during sorption of small quantities of vapors of monomers, this also testifies about high effectiveness of processes of transforming growing grafted chains into blocks, which fix macromolecules of PMMA, and confirms volumetric character of processes of synthesis of PAN and PVDKh even under conditions of

severe diffusion conditions of realization by its direct method. Basis  $\Delta P$  is formed not in the adsorptive layer on the external surface of film, but within. In the sample the concentration gradient of monomer, which decreases from the surface into the depth, is created. In the beginning of process the inner layers, not yet softened by monomer, serve as the base, which retains the sizes/dimensions of sample; subsequently the grafted structures of PAN and PVDKh reinforce material and they prevent the relaxation of the oriented structures of PMMA during the heating and in the solvents. Under the given conditions the layers with a thickness of  $20-30\mu$  effectively are cross-linked; therefore in samples No's 2 and 4 ( $d=170\mu$ ) middle layers are soluble, and the outer grafted layers (samples No's 3, 5) are separated/liberated.

Laminated structure of samples No's 2, 4 reflects their DIN [expansion unknown] (Fig. 1), form of which is obtained as if imposition of DIN prototype and grafted throughout entire volume sample. The isolated by solvent layers of PMMA with CCPAN (sample No 5) retain orientation, but with PVDKh (sample No 3) - no. Lower effectiveness of PVDKh, possibly, is connected with the formation of larger/coarser structures of PVDKh, i.e., with the smaller concentration of blocks in the sample with  $\Delta P$ , commensurate with PAN. Analogously, it is possible to explain the capacity of PVDKh to form the oriented structures ( $R=1.4$ ) during synthesis PAN  $R=1.05$  under the same conditions).

Strength ( $\sigma_p'$ ) of grafted samples No's 2, 4, 7 is not lower than initial, although they contain significant quantities of PAN and PVDKh ( $\Delta P$ ) and breakdown elongations ( $\epsilon_p$ ) of them somewhat above. After heating (15 min., 160°) the grafted samples to the greater degree retain  $\sigma_p'$  and  $\epsilon_p$  than initial (especially in grafted evenly by the volume of the sample, which retains a certain strength even at 160°). This also confirms conclusion about formation in PMMA of the effective cross-linkings, which raise the heat capacity of material.

Table 1. Structure and the property of the grafted oriented films of polymethyl methacrylate.

№	(1) Образец				(2) Термомеханические свойства						
	d, $\mu$	ПП	$\Delta P$ , %	R	$T_H$ , °C	$T_{max}$ , °C	$T_p$ , °C	$T_p$ , °C	$\sigma_{max}$ , кг/мм <sup>2</sup>	(3) ПММА*, $\sigma_{max}$ кг/мм <sup>2</sup>	$E_{co}$ , кг/см <sup>2</sup>
1	70	—	—	1,3	63	83	—	133	0,200	0,200	0
2	170	ПВДХ	28	1,4	59	80	—	218	0,252	0,297	—
3	20	ПВДХ	—	1,4	—	—	—	—	0	—	—
4	170	ПАН	18	1,04	64	90	—	245	0,235	0,278	—
5	30	ПАН	—	—	65	83	150	220	0,092	—	—
6	70	ПАН	75	1,06	63	83	155	—	0,135	0,238	0,45
7	80	ПАН	9	—	75	100	—	190	0,095	0,098	0,05

(5) Механические свойства			
$\sigma_p$ , %	$\sigma_p$ , кг/мм <sup>2</sup>	$(\sigma_p)_{T/\sigma_p}$	$(\epsilon_p)_{T/\epsilon_p}$
20	5,5	0,27	0,40
70	8,8	0,40	0,57
11	2,0	—	—
35	5,6	1,4	1,4
12,5	4,5	—	—
30 (40)	5,6 (0,035)	0,66	0,5

Key: (1). Sample. (2). Thermomechanical properties. (3). kg/mm<sup>2</sup>. (4). kgf/cm<sup>2</sup>. (5). Mechanical properties.

FOOTNOTE 1.  $\frac{ПММА}{\sigma_{max}}$  - relation  $\sigma_{max}$  and to fraction/portion of PMMA in the grafted sample.

2. Equilibrium modulus at 180°.

3.  $(\sigma_p)_{T/(\epsilon_p)_{T}}$  - tensile strength and elongation after aging of samples at 160° of 15 min. In the parentheses the results of strength tests at 160°. ENDFOOTNOTE.

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During heating to 300-400° are cross-linked with PAN samples they change chemical nature, which evidently from results of study (Fig. 1) by method of DTA of PMMA is destroyed and sample, without changing form, is converted into polyene cross-linked thermoresistant material,

i.e., in principle prototype of PMMA is matrix/die for direct synthesis in it of polymeric material of another structure.

Thus, study of properties and structure of grafted oriented amorphous PMMA made it possible to demonstrate validity of proposed into [3] mechanism of cross-linking polymers by blocks of grafted chains. Furthermore, is established/installed the possibility of the effective cross-linking of the oriented amorphous materials by the graft polymers, whose structures can fulfill the functions of crystallites, which offers essentially new possibilities of the modification of the physicochemical and mechanical properties of the oriented polymeric amorphous materials with the synthesis in them of the "crystallites" of the graft polymers of necessary nature with the optimal according to the sizes and by the volume of materials.



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